



TOPIC 2. STRUCTURE OF MATERIALS III

Topic 2.3:

- **Crystalline defects.**
- **Solid solutions.**

PERFECT AND IMPERFECT CRYSTALS

Perfect crystal: All the atoms are in the ideal lattice positions (only true at 0 K).

Real crystal: atoms vibrate

- There positions that are not occupied (vacancies)
- atoms displaced from ideal positions
- There are defects that modify the properties

PERFECT AND IMPERFECT CRYSTALS

Thermodynamics $\Rightarrow$ Justify $\exists$ of defects, as $\downarrow \Delta G_{\text{crystal}}$
 Creation of a defect $\Rightarrow \Delta H > 0$ and $\Delta S > 0$

Configuration Entropy $\Rightarrow S = k_B \ln W$

k_B = Boltzmann constant = $1.380622 \cdot 10^{-23} \text{ JK}^{-1}$

W = way of distributing n defects in N possible positions arbitrarily

In 1 mol of C^+ there are N_A possibilities to create 1 vacancy

$W \propto 10^{23}$

$$W = \frac{N!}{(N-n)!n!}$$

Vacancy or defect:

$$\Delta H < T\Delta S$$

$$\Delta G = \Delta H - T\Delta S < 0$$

When $\uparrow T \Rightarrow \uparrow$ [defects]

DEFECTS TYPES (according to dimensions)

1. Point :

Vacancies
Interstitials
Substitutional
Schottky
Frenkel
Order-Disorder

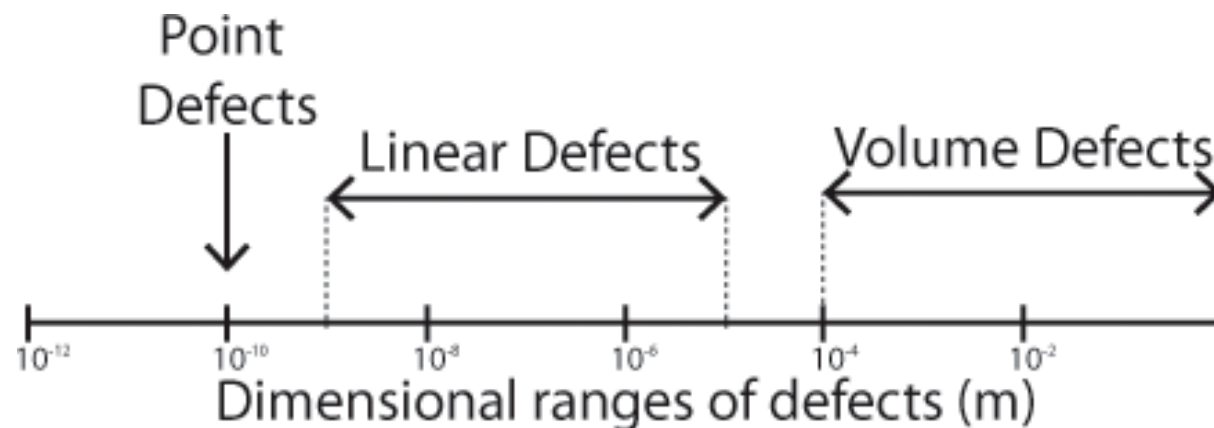
2. Linear: dislocations:

Edge
Screw
Mixed

3. Complex: "clusters"

4. Planar or Extended

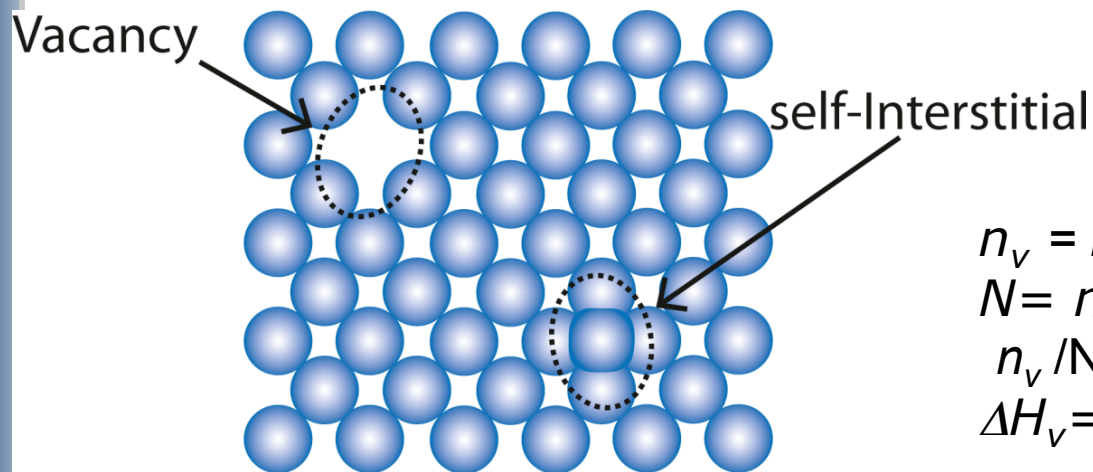
External Surfaces
Grain Boundaries
Twin Boundaries
Stacking Faults



POINT DEFECTS: VACANCIES AND INTERSTITIALS

Vacancy: When an atom is missing from a position where it should be.

Interstitial: An atom from the crystal occupies the place of an interstitial (*self-interstitial* or *interstitialcy*)



$$n_v = N \exp\left(\frac{-\Delta H_v}{RT}\right)$$

$n_v = n^0$ vacancies per cm^3

$N = n^0$ of lattice points per cm^3

$n_v / N =$ fraction of vacancies

$\Delta H_v =$ Formation energy (J.mol^{-1})

vacancies concentration at equilibrium $\frac{N_v}{N} \approx 10^{-4} \text{ max}$

They are produced : $\Rightarrow$ during solidification (impurities, alloys)
 $\Rightarrow$ by particle bombardment with $\uparrow E$
 $\Rightarrow$ during plastic deformation (processing)
 $\Rightarrow$ when $\uparrow T \Rightarrow \uparrow$ thermal vibrations $\Rightarrow \uparrow n_v$

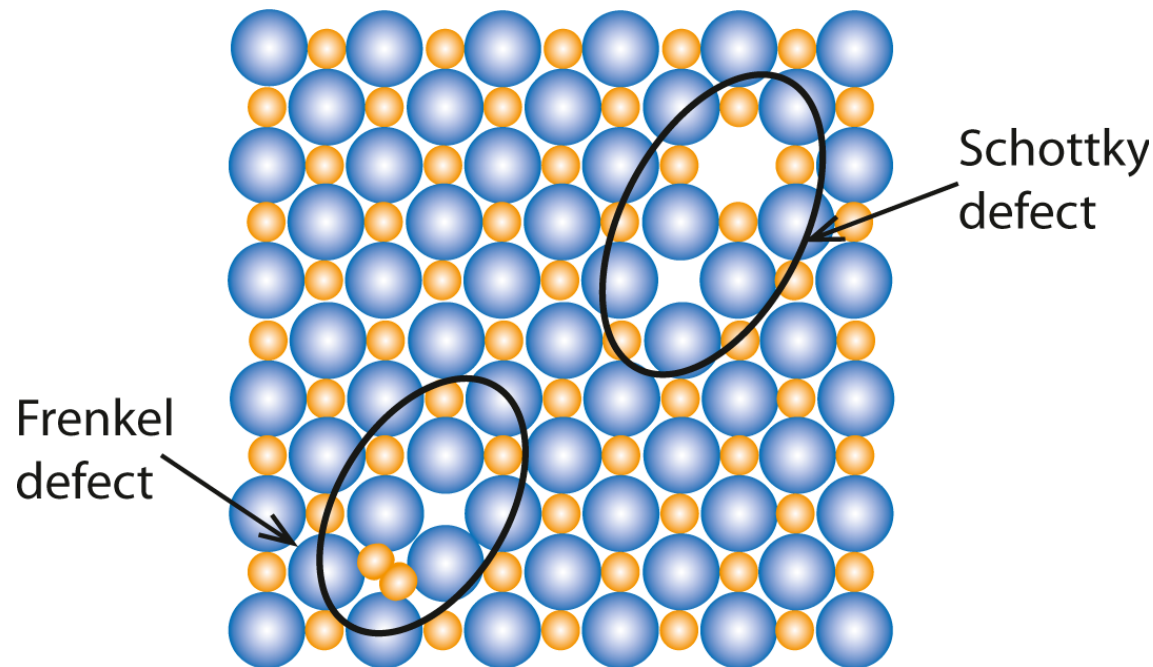
POINT DEFECTS: SCHOTTKY AND FRENKEL

DEFECTS IN **IONIC CRYSTALS**

Schottky Defect: Vacancy in an ionic crystal.

In order to maintain neutrality $\Rightarrow$ Anionic and cationic vacancies appear

Frenkel Defect: migration of an ion from its normal position to an interstitial position (combination vacancy-interstitial)



POINT DEFECTS: SCHOTTKY AND FRENKEL

Nº of Schottky defects: n_s

$$n_s = N \exp\left(-\frac{\Delta H_s}{2RT}\right)$$

Nº of Frenkel defects : n_F

$$n_F = \sqrt{NN_i} \exp\left(-\frac{\Delta H_F}{2RT}\right)$$

$\Delta H_S = E$ for Schottky defect creation

$\Delta H_F = E$ for Frenkel defect creation

$N = n^0$ of lattice positions

$N_i = n^0$ of interstitial positions

$T =$ Temperature (K)

In a crystal $\Delta H_S \neq \Delta H_F \Rightarrow$ the defect with the lowest ΔH will form

n^0 defects increases with temperature

POINT DEFECTS: SCHOTTKY AND FRENKEL

	Compound	ΔH (eV)	ΔH (10^{-19} J)
Schottky defects	MgO	10.57	6.60
	CaO	9.77	6.10
	LiF	3.75	2.34
	LiCl	3.40	2.12
	NaCl	3.69	2.30
	KCl	3.62	2.26
Frenkel defects	ZrO ₂	6.57	4.10
	CaF ₂	4.49	2.80
	SrF ₂	1.12	0.70
	AgCl	2.56	1.60
	AgBr	1.92	1.20

NaCl ($T_f=801\text{ }^\circ\text{C}$) $\Delta H_s = 3.69 \times 10^{-19}\text{ J}$
 $T=300\text{K}$ $n_s = 2.64 \times 10^4$ vacancies/mol
 $T=1000\text{K}$ $n_s = 9.38 \times 10^{17}$ vacancies/mol

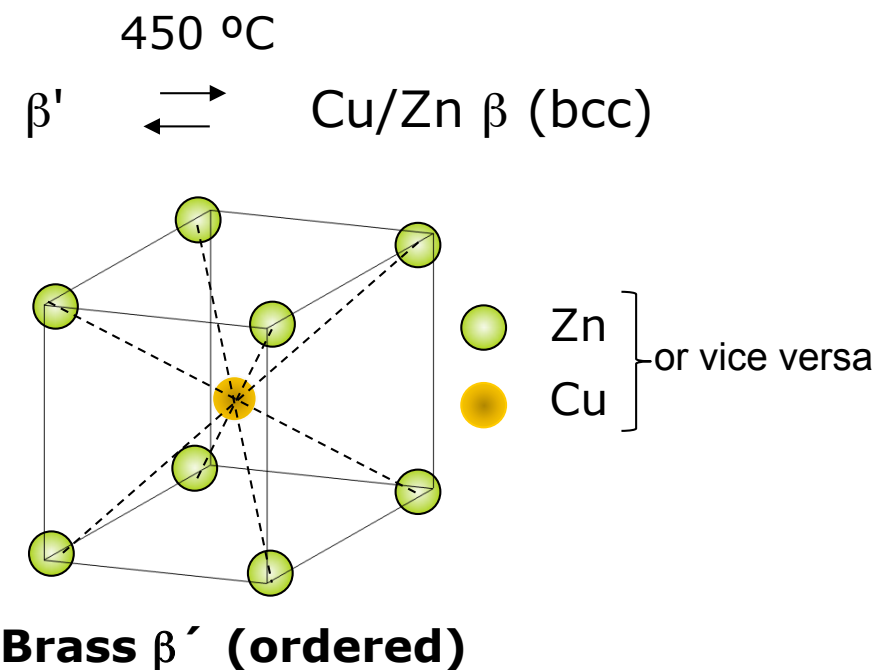
MgO $\Delta H_s = 10.57 \times 10^{-19}\text{ J}$
 $T=300\text{K}$ $n_s = 2.12 \times 10^{-32}$ vacancies/mol
 $T=1000\text{K}$ $n_s = 1.39 \times 10^7$ vacancies/mol

$\Delta H_s(\text{MgO}) > \Delta H_s(\text{NaCl})$
 $\Rightarrow n_s$ low, more difficult to create vacancies.

POINT DEFECTS: ORDER-DISORDER IN SOLID SOLUTIONS

Order-disorder phenomena (in substitutional solid solutions):

Atoms of one sub-lattice occupy positions corresponding to the other and vice versa (metallic alloys). *Solids with elements that have similar electronegativity*

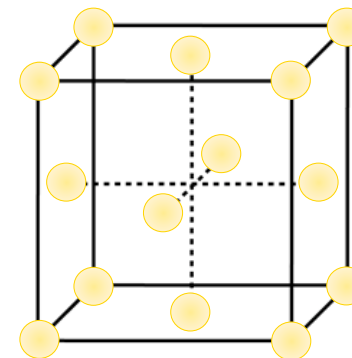


Brass β' : Cu centre Zn corners

Brass β : Cu and Zn arbitrarily distributed in BCC

$T > 390 \text{ } ^\circ\text{C}$

Disordered



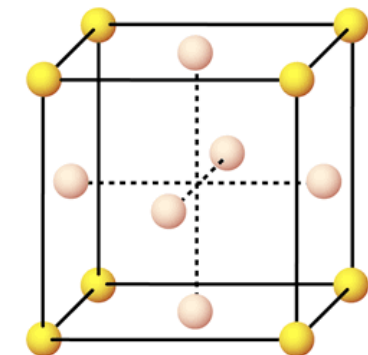
“average” gold-copper atom

copper atom

gold atom

$T < 390 \text{ } ^\circ\text{C}$

Ordered



Cu-Au Alloys

LINEAR DEFECTS : DISLOCATIONS

DISLOCATIONS

They are defects that produce distortion in the lattice situated around a line.

Characteristics: they can be displaced in the interior of a crystal by applying relatively low forces and can produce a complete displacement over crystalline planes.

Explain: - $E_{\text{theoretical}}$ (Young modulus) > $E_{\text{experimental}}$
- Plastic deformation in metals (workability, ductility)

Formation:

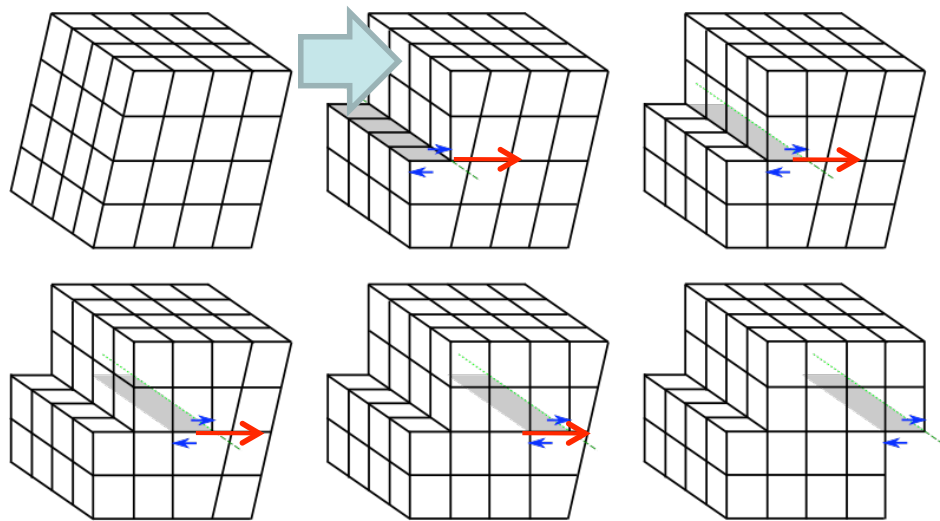
- during solidification
- through plastic or permanent deformation of crystal
- through concentration of vacancies
- through atomic disarrangements in solid solutions

Types:

- edge dislocation, or Taylor dislocation
- screw dislocation, or Burgers dislocation
- mixed dislocation

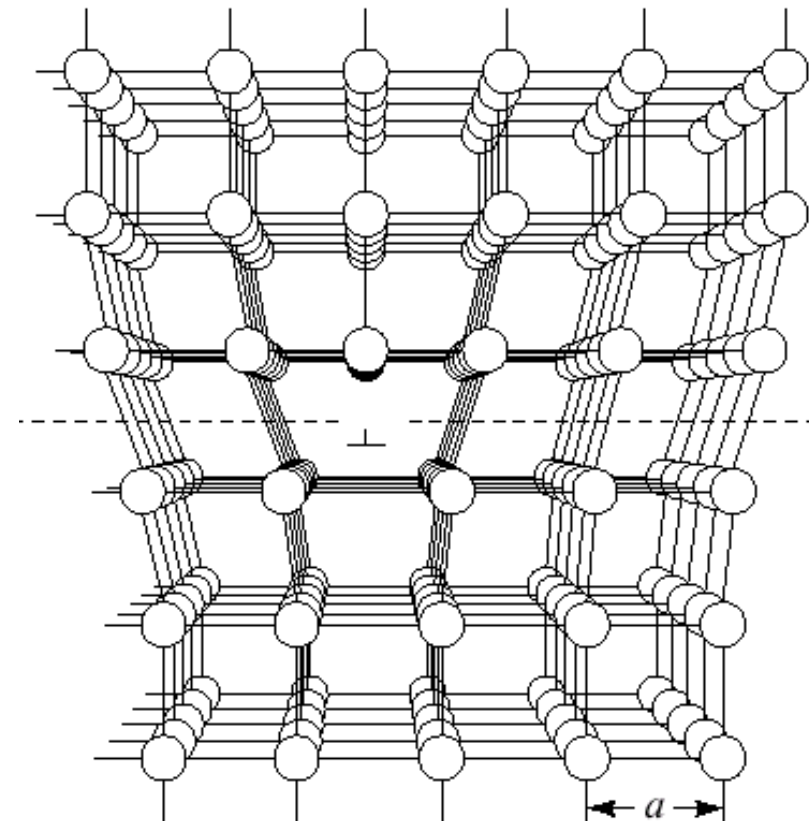
EDGE DISLOCATIONS (or TAYLOR)

⇒ Geometric modification of the lattice: equivalent to an extra plane of atoms ⇒ Energetic modification (they store energy)



Direction of motion

http://commons.wikimedia.org/wiki/File:Dislocation_coin_et_deformation_3d.svg



Dislocation line.- $\perp$ point A
Slipping plane.- line of points

William D. Callister, Jr., Materials Science and Engineering : An Introduction, John Wiley & Sons, Inc.

EDGE DISLOCATIONS (or TAYLOR)

* Dislocation characterization: Burgers Vector ($\mathbf{b}$)

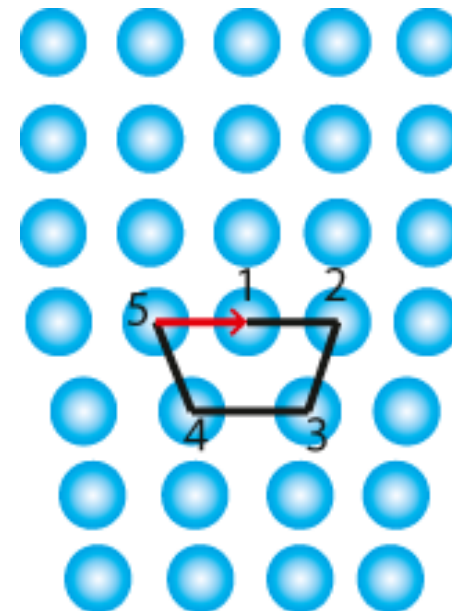
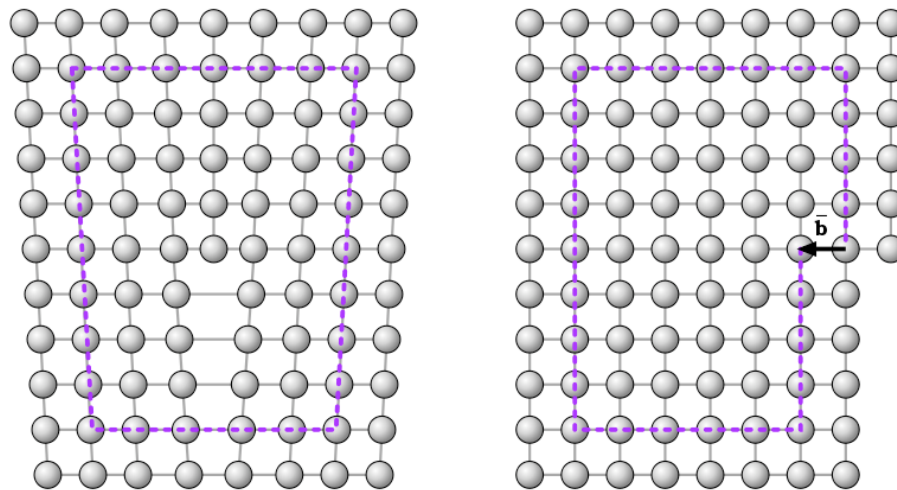
Burgers Vector: magnitude and direction of the lattice distortion associated with the dislocation

Magnitude: distance 1-5

Direction: 1-5 or 5-1

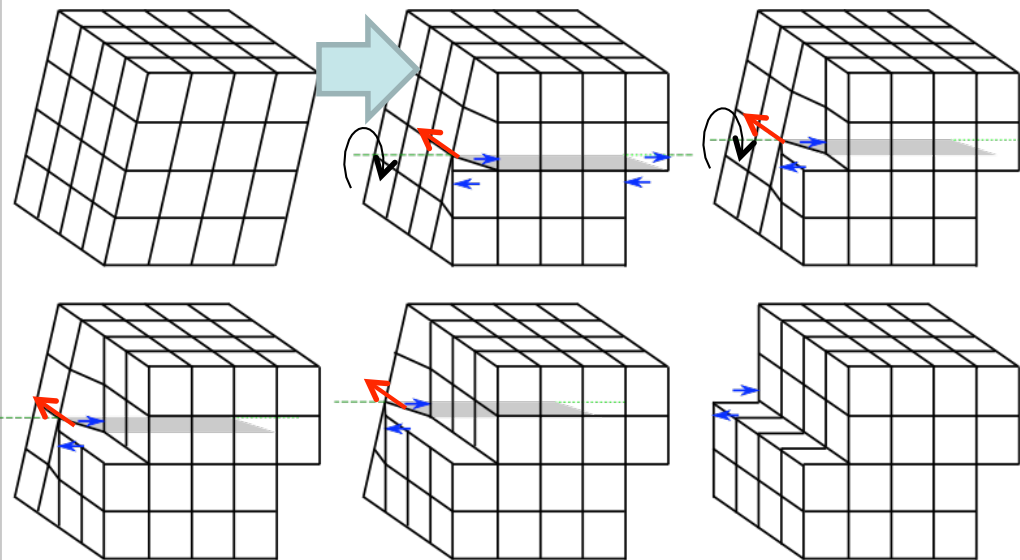
$\mathbf{b} \perp$ to dislocation line

$\mathbf{b} \parallel$ to the direction movement



SCREW DISLOCATIONS (or BURGER)

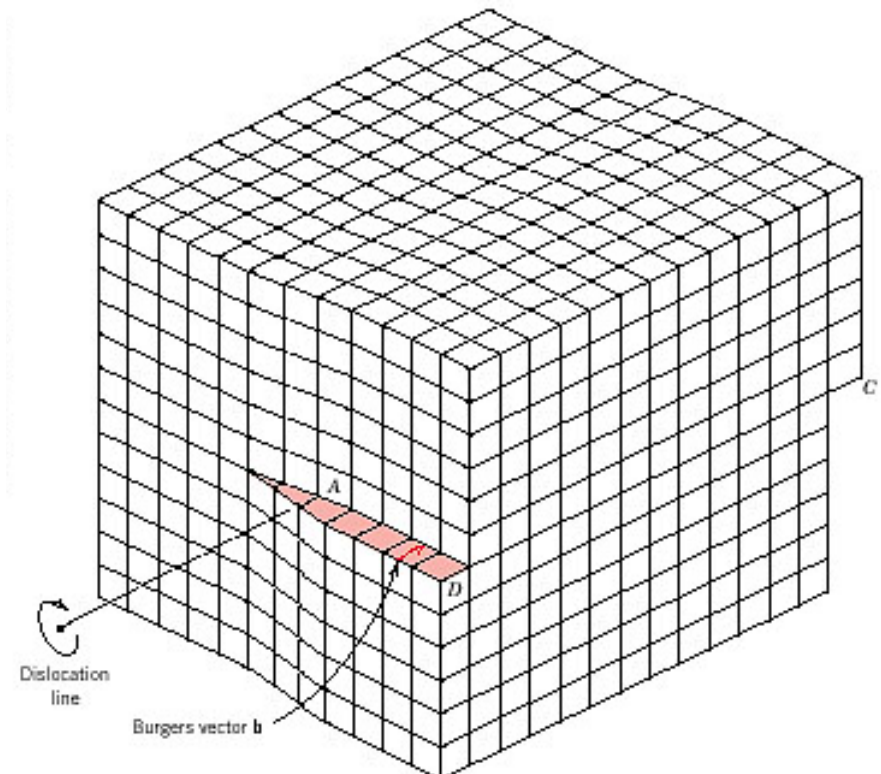
⇒ Locally curves some atom lines.
The effect is as if a shear stress is applied to produce a distortion



Direction of motion

http://commons.wikimedia.org/wiki/File:Dislocation_vis_et_deformation_3d.svg

We can move from the inferior to the superior plane through the dislocation line.



William D. Callister, Jr., Materials Science and Engineering : An Introduction, John Wiley & Sons, Inc.

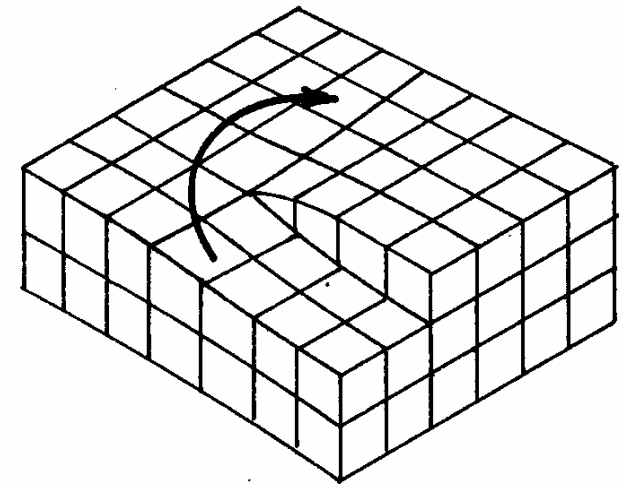
SCREW DISLOCATIONS (or BURGER)

* Burgers Vectors in screw dislocations:

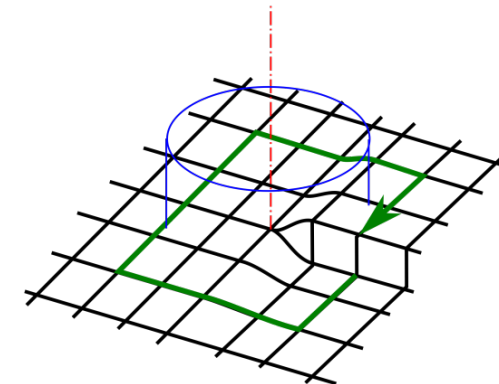
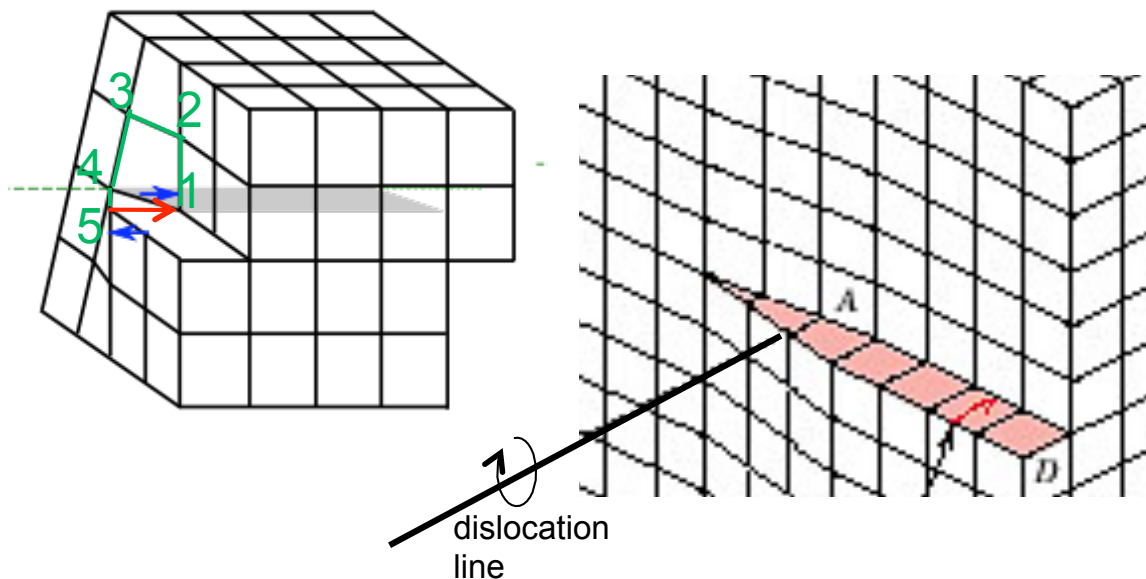
The magnitude and distance from 1-5 define **b**

b || to dislocation line SS'

b $\perp$ to the direction of movement



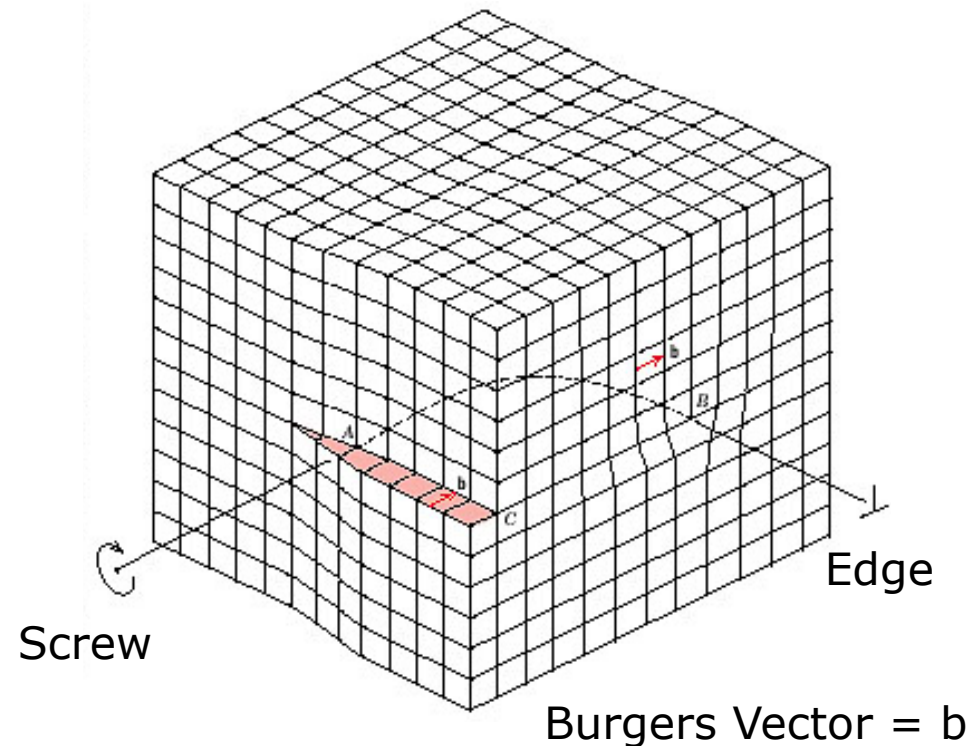
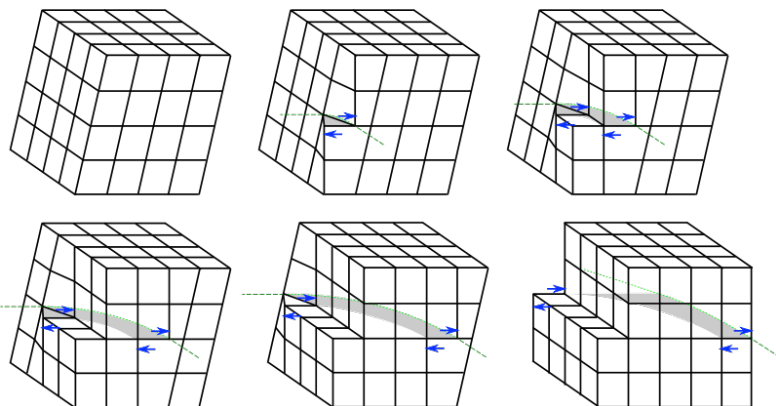
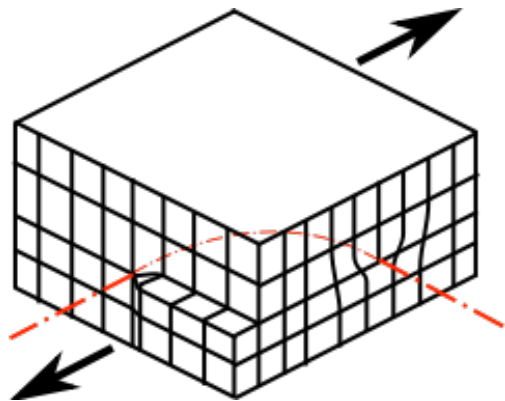
We can trace a screw (spiral ramp) around the dislocation.



http://commons.wikimedia.org/wiki/File:Dislocation_h%C3%A9lico%C3%AFdale.png
http://commons.wikimedia.org/wiki/File:Dislocation_vis_vue_en_perspective_et_coeur.svg

MIXED DISLOCATIONS

If there are many dislocations $\Rightarrow$ they are mixed (**MIXED dislocations**)



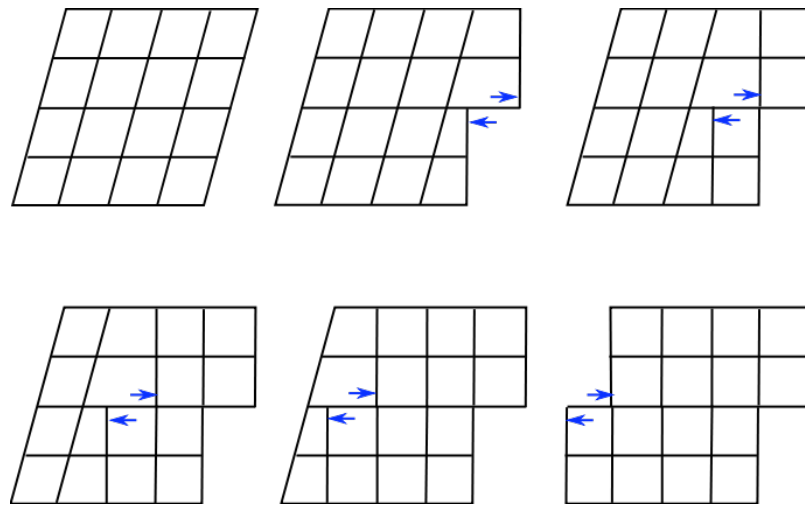
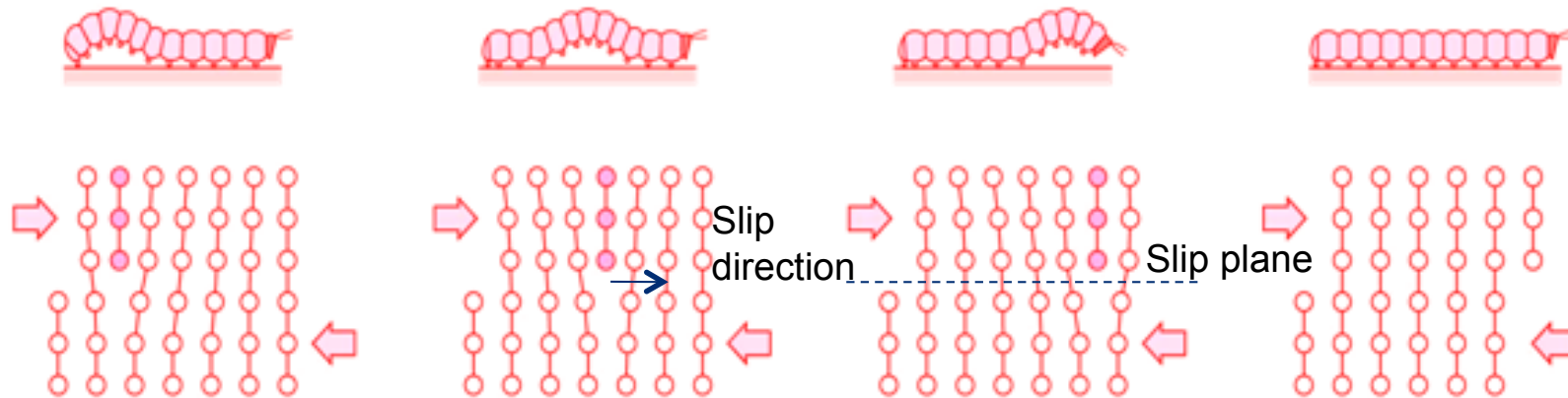
The Burgers vector will be the same at all points along its line even though the dislocation might change direction and nature within a crystal.

http://commons.wikimedia.org/wiki/File:Dislocation_mixte_perspective_iso.svg
http://commons.wikimedia.org/wiki/File:Dislocation_mixte_et_deformation_3d.svg

William D. Callister, Jr., Materials Science and Engineering : An Introduction,
 John Wiley & Sons, Inc.

DISLOCATIONS MOVEMENT

Dislocation movement (edge and screw) when a shear stress is applied



slip direction: the direction in which the dislocation moves

slip plane: the plane in which the dislocation moves

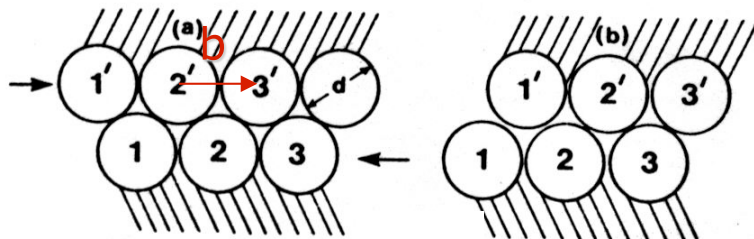
The combination of the slip direction and the slip plane is the **slip system**

PLASTIC DEFORMATION : SLIPPING OF DISLOCATIONS

Dislocations movement parallel to the directions of maximum packing $\Rightarrow$ Lower distance between them $\Rightarrow$ + easy slipping

$$E \text{ used to move a dislocation} = E \propto |b|^2$$

Compact Direction

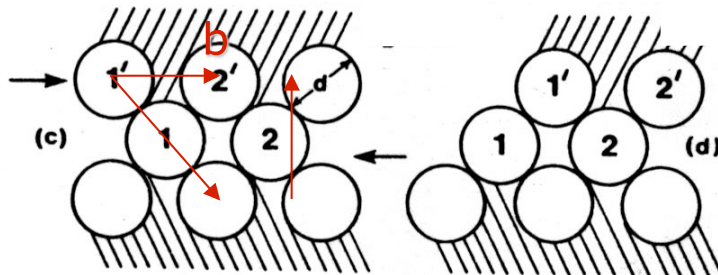


For metallic materials, the burgers vector for a dislocation will be the magnitude equal to the interatomic spacing.

$$|b| = 2R$$

$$E_b \propto 4R^2$$

NON Compact Direction



$$b^2 + b^2 = (4R)^2 \Rightarrow b = 2\sqrt{2} \cdot R$$

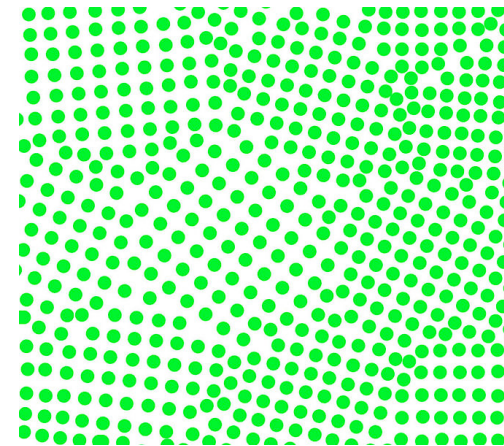
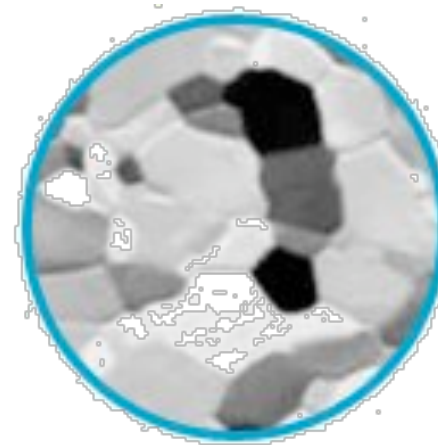
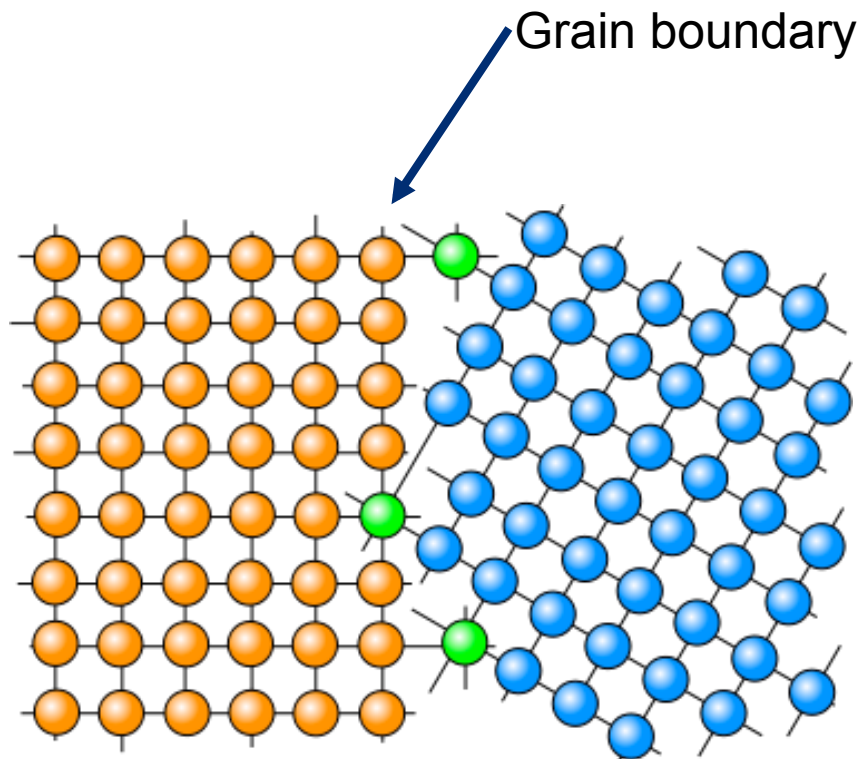
$$E_b \propto 8R^2$$

A.R. West. "Solid State Chemistry and its applications". Wiley.Chichester,1992

PLANAR DEFECTS: GRAIN BOUNDARIES

They separate crystals with different orientation .

Formation: during solidification: $\neq$ crystals are formed when $\neq$ nuclei grow simultaneously.



They limit dislocations movement in the material

http://commons.wikimedia.org/wiki/File:Joint_de_grain_reseau_coincidence.svg

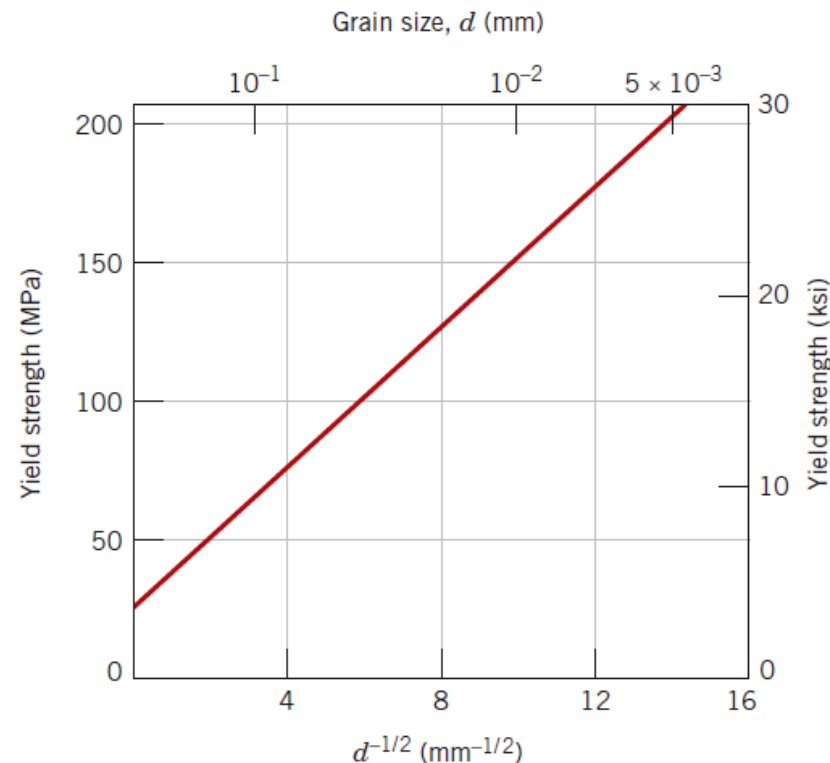
PLANAR DEFECTS : GRAIN BOUNDARIES

Boundary width: 2-5 interatomic distances

Very energetic zones (atomic packing < less than in interior):

- Solid state reactions
- ↑ Atomic diffusion

Materials properties = f (grain size)



**When grain size ↓
⇒ Mechanical resistance ↑
Dislocation movement is
limited when there are
many grain boundaries.**

William D. Callister, Jr., Materials Science and Engineering : An Introduction, John Wiley & Sons, Inc.

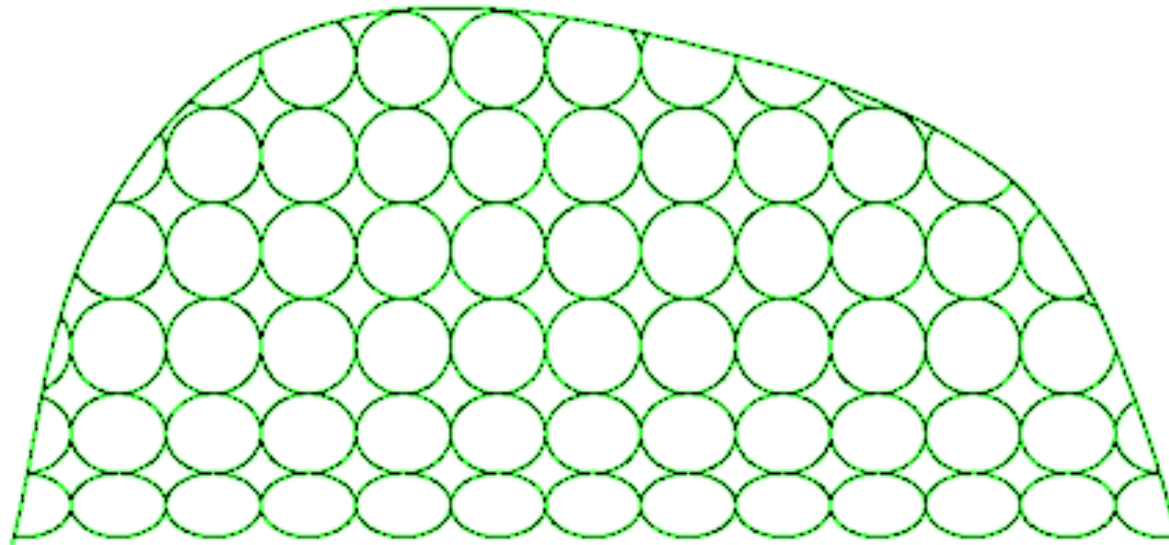
PLANAR DEFECTS: EXTERNAL SURFACES

EXTERNAL SURFACE

It is the end of the crystal or grain structure

Coordination numbers at the surface < at the interior of the crystal $\Rightarrow$

$$E_{\text{surface}} > E_{\text{interior}}$$



PLANAR DEFECTS : STACKING FAULTS

STACKING FAULTS

In fcc and hcp structures

hcp

fcc

perfect lattice :

...ABABAB...

...ABCABCABC...

Stacking fault:

...ABABCAB...

...ABCABABC...

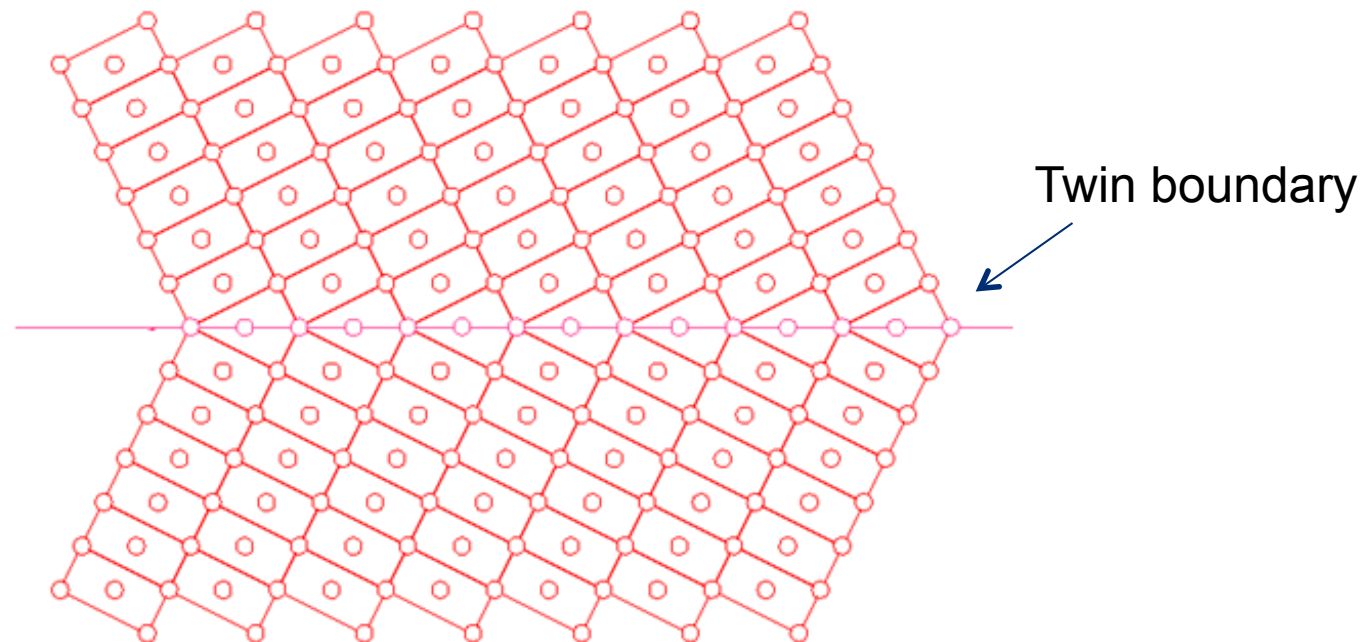
PLANAR DEFECTS : TWIN BOUNDARIES

Twin boundary:

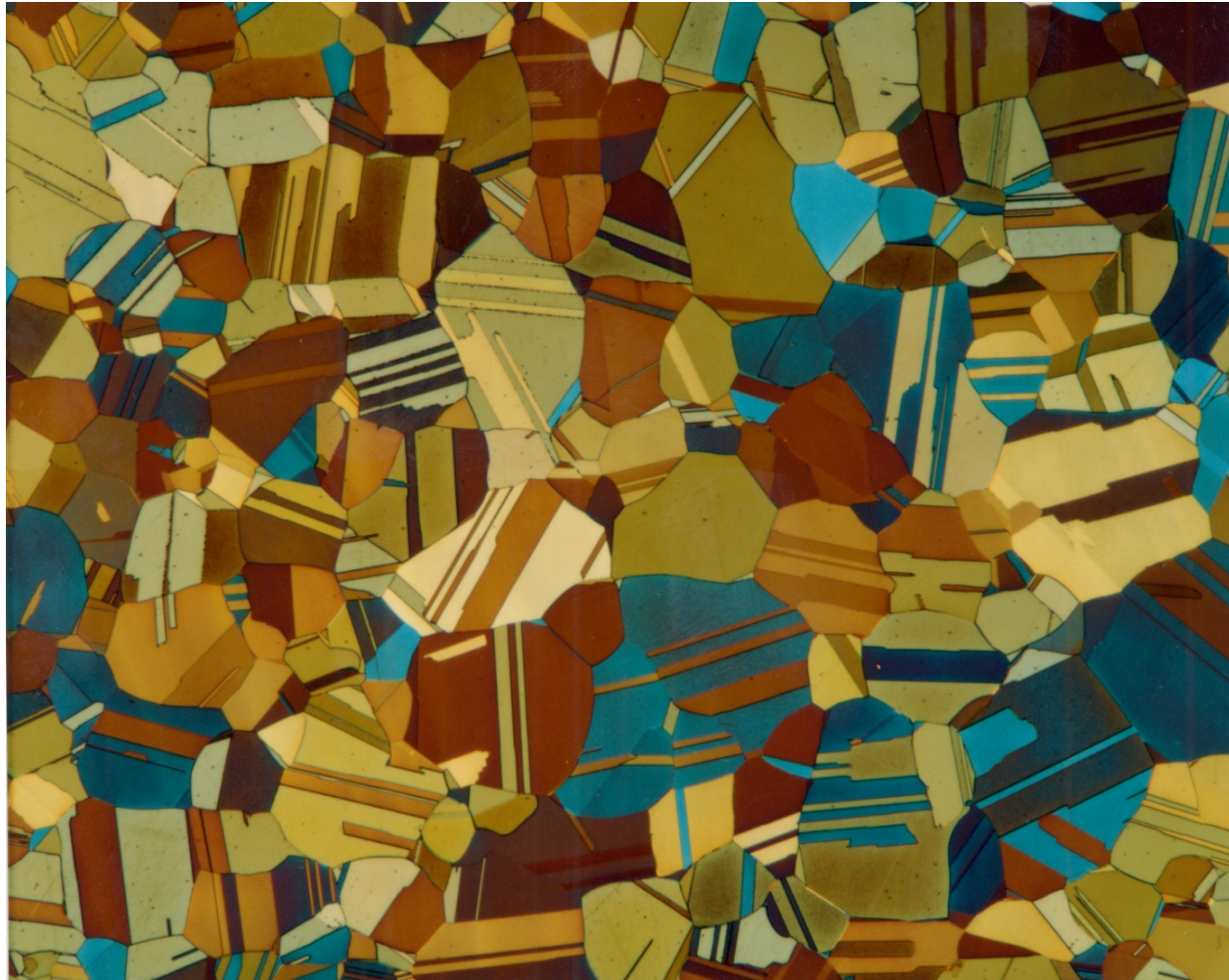
Boundary that separates two parts of a crystal with a small difference in the orientation (misorientation) of the planes (crystal twinning). Highly symmetrical interface, often with one crystal the mirror image of the other

Formation: Deformation process or during thermal treatments

*Produce $\uparrow$ mechanical resistance $\Rightarrow$ make difficult dislocation gliding



PLANAR DEFECTS: TWIN BOUNDARIES



Brass (70 Cu / 30 Zn)

SOLID SOLUTION

Formation of a **SOLID SOLUTION**

Def. "Solid that has 2 or more elements dispersed as atoms in a structure that has only one phase "

"Phase of variable composition"

When we add another element to a material in the liquid state

FORMATION of a new **PHASE**

SUBSTITUTIONAL

solute or impurity atoms replace or substitute the host atoms

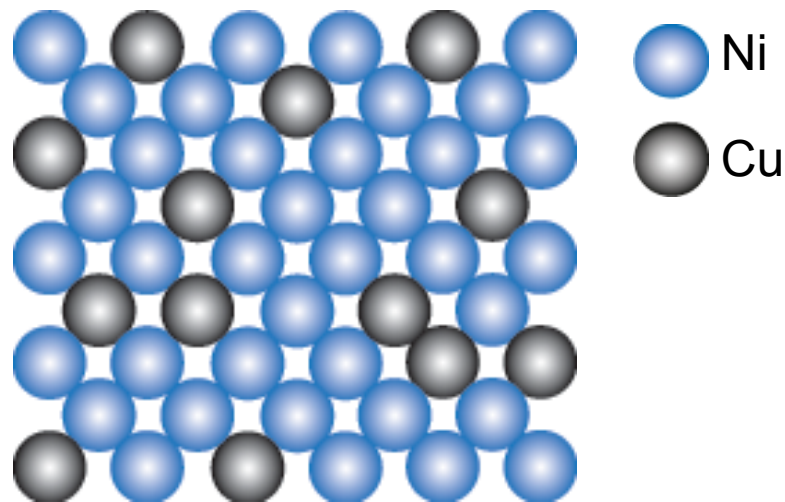
INTERSTITIAL,

elements with small radius in octahedral or tetrahedral sites (H, C, N, B y N)

SOLID SOLUTION

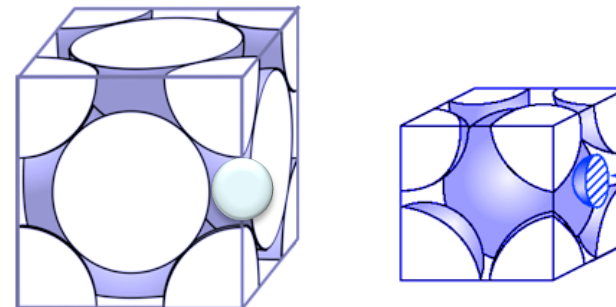
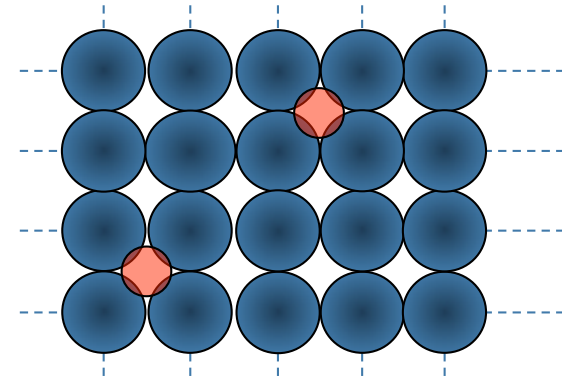
A) Substitutional solid solution

$T > T_f$ during cooling
 70g Cu (fcc) + 30 g Ni (fcc) in
 liquid state → new structure with
a intermediate between Cu and Ni



B) Interstitial solid solution

Small atoms located in octahedral
 sites or tetrahedral sites



SOLID SOLUTION : HUME-ROTHERY RULES

Hume-Rothery rules for the formation of substitutional solid solutions :

- 1) The same crystal structure
- 2) Atoms or ions with **similar radius**
 - For ceramics difference $< 30\%$
 - For metals difference $< 15\%$
 - For differences $> 15\%$ limited solubility ($< 1\%$)
- 5) They must have similar **electronegativities**
- 6) **Valance:** In ceramics: They must have the same valance.

If all conditions are fulfilled $\Rightarrow$ we do not necessarily have total solubility_

-If one or more conditions are not fulfilled $\Rightarrow$ partial solubility